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Characterisation of Al corrosion and its impact on the mechanical performance of composite cement wasteforms by the acoustic emission technique

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Abstract

In this study acoustic emission (AE) non-destructive method was used to evaluate the mechanical performance of cementitious wasteforms with encapsulated Al waste. AE waves generated as a result of Al corrosion in small-size blast furnace slag/ordinary Portland cement wasteforms were recorded and analysed. The basic principles of the conventional parameter-based AE approach and signal-based analysis were combined to establish a relationship between recorded AE signals and different interactions between the Al and the encapsulating cement matrix. The AE technique was shown as a potential and valuable tool for a new area of application related to monitoring and inspection of the mechanical stability of cementitious wasteforms with encapsulated metallic wastes such as Al. © 2008 Elsevier B.V. All rights reserved.

1. Introduction

All processes involved in the nuclear energy production are associated with generation of gaseous, liquid and solid wastes with various concentrations of radioactive elements [1]. In the UK these are classified according to their nature and level of radioactivity into very low level waste (VLLW), low level waste (LLW), intermediate level waste (ILW) and high level waste (HLW) [1,2]. VLLW has a negligible radiological hazard and is safety disposed of at landfill sites. The liquid and solid LLW and ILW arise mainly from the operation and decommissioning of power stations and spent nuclear fuel (SNF) reprocessing. Most of the low and intermediate level wastes (LILW) have been immobilised in cements and sent to the UK storage facilities near Drigg in Cumbria. The mechanical stability along with radiation and chemical durability of the produced wasteforms are the most important properties required for their

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safe storage, transportation and disposal. According to the latest UK Radioactive Waste Inventory report 31557 cementitious packages immobilising solid and liquid ILW were produced up to 1 of April 2004 and the forecast for encapsulation of future arisings determines a number of 177 000 cement-based wasteforms to be produced [2]. The dominant ILW encapsulated in cements are ion exchange resins, sludges from treated radioactive liquid effluents and metallic debris such as SNF claddings and reactor components [3]. These wastes have been progressively encapsulated in composite cements [3,4] sealed in 500-l stainless steel drums or for large items within high capacity concrete or steel containers [5]. Over a number of years baseline data were accumulated for various waste streams immobilised in blast furnace slag (BFS)/ordinary Portland cement (OPC) and pulverised fly ash (PFA)/OPC composite cements. The physical and chemical properties of the cementitious wasteforms such as setting time, temperature elevation during early stage of hydration, permeability, chemical durability, tensile strength and dimensional stability have been studied [3]. However, the mechanical behavior of the cementitious wasteforms during long

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periods of time (tens and hundreds of years) is difficult to be predicted and can be substantially affected by gas generation and corrosion of wasteforms constituents [3]. An important issue is the degradation of the cementitious wasteforms caused by corrosion of encapsulated metals such as Al, Mg and their alloy, termed Magnox, in the high pH cement environment (~12.5 for OPC) associated with hydrogen gas generation and deposition of an expansive layer of corrosion products [6,7]. This is a major concern for the safe storage of more than 23 000 cementitious packages with encapsulated Magnox and Al which have been or will be produced only by Sellafield encapsulation plants in the UK [2]. These packages are in interim storage for minimum of 50 years before final disposal. Therefore nondestructive methods for continuous monitoring and inspection of the mechanical performance of the cementitious wasteforms, such as acoustic emission (AE) testing, would be of significant benefit for assessment of their mechanical stability and final acceptability for handling, transport and disposal at the end of the interim storage.

2. AE technique applied for structures immobilising nuclear wastes

AE is a natural occurring phenomenon associated with sudden release of elastic energy propagating in the form of transient waves caused by mechanical deformation, dislocation movement, phase transformation or other irreversible mechanical changes within different materials [8]. The largest scale AE sources are the seismic events whereas processes that still generate detectable AE are microscopic defect movements of the order of a few picometers in a stressed structure [9]. The collection and processing of AE waves was developed and established as a non-destructive testing and evaluation (NDT&E) method particularly shown as a powerful tool for characterisation of failure mechanisms and damage assessment in various materials and structures [9,10].

The AE technique offers several advantages which make it a potentially suitable tool for non-destructive testing and inspection of nuclear waste packages. These are related to the high sensitivity of the method to micro and macromechanical events within the structures providing an early indication for the damage development in situ, the possibility to be used as a passive testing method without additional stimuli to be applied and, in theory, unlimited duration of monitoring. Moreover, the AE practice has shown that testing can be conducted on structures with different composition, shape and volume and successfully used for localisation of zones of damage [9-11]. The AE technique also uses relatively simple but effective means of monitoring the time progress development of different failure mechanisms. It employs approaches to differentiate them by AE data analysis which can almost instantly alarm for un-controlled deterioration or human intrusion. Nevertheless, the application of the AE technique is associated with a number of issues. These are related to the diversified

AE signal characteristics determined by the type of defects, deterioration mechanisms, local stress distribution, mechanical properties and geometry of the structures under monitoring, background noise, experimental conditions, e.g., high temperature, aggressive solution or radioactivity, and finally location and intrinsic properties of the sensors employed [9–11]. These factors determine the complexity of the experimental setup and the repeatability of the results obtained. Therefore a strict correlation between dimensional indications obtained by the detected AE waves and the nature of the failure process can be determined based on relatively complicate data interpretation methods including advanced signal processing, e.g., wavelet and moment tensor analysis [11-13]. However, the information gained from defects growth, onset of new microfracture patterns and their dynamics, especially in cement-based structures, supply useful elements for a diagnosis of the degradation phenomena in progress [14–20].

Apart from our previous work [21–23] there are few studies on AE as a NDT&E technique applied for nuclear wasteforms. Belov and Aloy [24] reported that AE signals can be detected, distinguished from the background noise and associated with defects formation during fabrication of glass and ceramic samples used for immobilisation of nuclear wastes. Even though different AE methods have been extensively developed and optimised for damage characterisation in cement-based structures for civil engineering applications at the authors' knowledge the AE technique has not been applied to monitor the mechanical stability of cementitious wasteforms; thus this a new area with a potential for future development. We recently reported the results obtained from the AE monitoring of a pure OPC structure with encapsulated Al used as a reference for further study on samples resembling real cementitious wasteforms with encapsulated Al [21–23]. Different methods for analysis of the collected AE data were applied in order to characterise the potential sources of AE within the cementitious system. It was shown in [21–23] that the AE signals recorded from the reference OPC structure with encapsulated Al can be classified and associated with different micro and macro scale mechanisms leading to damage development within the structure with the time progress of the Al corrosion process. The aim of this study was to extend the work, briefly reported in [21–23], on composite cement formulation used for production of real wasteforms for encapsulation of metallic radioactive wastes [1,3,4]. Herein the evaluation of the damage induced within the structures was performed using AE parameter and signalbased analysis.

3. Experimental procedure

3.1. Specimens

AE monitoring was conducted on composite cement samples with 70 wt% BFS replacement of OPC (BFS/OPC to mass ratio 7:3) used as a lower limit in the compo-

sition envelope for nuclear waste encapsulation applied by British Nuclear Group (BNG). The metal encapsulated was high purity (99.999%) Al. The OPC and BFS were supplied respectively by Castle Cement and Redcar Steel Works.

For the preparation of the laboratory scale units the OPC and BFS powders with water, in mass ratio 0.33, were mixed and then poured in 340 mL capacity plastic containers. Before to close the container with an airtight lid Al rod with a length between 35 and 50 mm and a diameter of 10 mm was centrally placed into the cement grout. Cylindrical cementitious samples were formed and cured at 20 °C and 95% relative humidity (RH) in an environmental chamber for different period of time – from 3 days to more than 3 years. For the purpose of this study the corrosion of Al in cementitious samples, cured for more than 6 months, was accelerated by adding 50 mL of deionized water in the plastic containers. The latter was done in order to simulate one of the possible scenario during the interim storage of the packaged wasteforms when water can penetrate and interact with the encapsulating cementitious matrix and the waste [1]. In this paper the results from the AE monitoring of a BFS/OPC sample cured for more than 6 months are presented and discussed.

3.2. AE data acquisition

The AE experimental setup, shown in Fig. 1, and layout for data acquisition were as described in [21-23]. A piezoelectric AE transducer (or sensor) with a diameter of 17 mm was attached to the bottom of the container with the cement-based sample via a thin layer of grease used as acoustic couplant. Although the quality of the sensor coupling affects the reproducibility of the characteristics of the detected AE waves all experiments on the cementitious samples were conducted by the same operator following the same procedure minimizing random effects. A wideband transducer, type WD, calibrated and supplied by Physical Acoustics Corporation (PAC) has been chosen due to its high sensitivity in a broad range of frequencies between 100 and 1000 kHz. The electrical signals (voltage) generated by the AE transducer were amplified by 40 dB and passed through a bandpass filter operating between 20 kHz and 3 MHz. Then each of the analog signals was measured, and if above the threshold level of 40 dB [25], was sampled and quantized by 18 bits analog-to-digital converter (ADC) with 5 MSPS sampling rate.

A hit driven data acquisition process with PCI-2 based AE system, supplied and calibrated by PAC, was used. When an AE wave was detected by the transducer attached to the sample, the generated electrical signal amplified by a preamplifier type 2/4/6 from PAC and its amplitude exceeded the threshold level data streaming was allowed. The data recorded on a PC hard drive in an ASCII file for each AE signal (or hit) consisted of a set of parameters such as duration, amplitude, counts, rise time and absolute (ABS) energy, defined in [21], and subsequently used for post-test analysis.

The waveform of each filtered and recorded AE signal was discretised in 3072 points stored in the PC memory. The pre-trigger time (in µs) used to record the signal waveform before the first threshold level crossing was set at 100 µs. Selective cross-plots such as hits amplitude versus duration and histograms, plotted by PAC AE win software package, provided real-time information for the AE from the sample under monitoring.

4. Results and analysis

4.1. Relationship between the time domain parameters of the recorded AE hits from the BFS/OPC sample with encapsulated Al and their potential sources due to the Al corrosion development

Signal parameter-based analysis is a conventional approach for AE sources characterisation and overall assessment of the mechanical performance of materials and structures [9,12]. It is suitable for less time and storage consuming data acquisition applications, e.g., monitoring and inspection of large concrete bridges, pressure vessels and oil tanks. However, the AE hits rate and parameters such as counts, amplitude and duration are one of the basic tools for AE analysis extensively used to describe and study the acoustic signature of cement-based structures in laboratory experiments under various conditions [14–20].

In the present study during the conducted AE monitoring of the cementitious sample with encapsulated Al the



Fig. 1. AE experimental setup used for monitoring of the BFS/OPC sample with encapsulated Al.

cumulative number of AE hits and associated ABS energy provided initial, real-time information for micromechanical events within the structure associated with the generation and release of acoustic waves. As it can be seen in Fig. 2 there were several stages in the AE history of the experiment with a discriminative rate (high or low) also reported from the AE monitoring of the reference OPC structure with encapsulated Al [21].

The first relatively long period of time (92 h) from the commencement of the experiment was associated with a very low AE activity consisting of 193 signals or 8.29% of the total number of the detected AE hits (Table 1). The sources of this AE can be related with the process of water infiltration through the porous zone in the Al corrosion products layer and re-initiation of the metal corrosion

leading to a stable growth of the pre-existing microcracks formed due to the hydrogen gas generation and release during the cement setting and curing [6,7].

After the first stage a new period characterised by an abrupt increase in the number of the recorded AE hits and their ABS energy was observed likely to indicate for localised microcracking and formation of critical size crack(s) within the structure [14–18,21]. This period with a duration of around 50 h (Fig. 2) has been followed by a time interval when the hits rate considerably decreased in comparison with the previous period denoted as 2a in Fig. 2. Similar results were reported to correspond to propagation of critical size cracks in cement-based materials during loading, debonding along the steel reinforcement in concrete or hydrogen gas release due to Al corrosion associated with

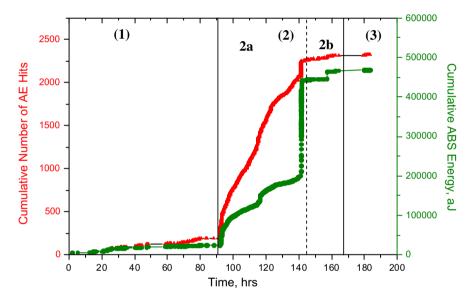


Fig. 2. Cumulative number of AE hits and associated ABS energy detected during the AE monitoring of the BFS/OPC samples with encapsulated Al.

Table 1
Time domain parameters of the hits recorded during the AE monitoring of the BFS/OPC sample with encapsulated Al

Time period	Hits duration, (μs)	Amplitude peak, (dB)	Amplitude range, (dB)	Max of counts/hits	Calculated average counts/ hits	$N_{ m hits}$	$N_{ m counts}$	$\%$ of total $N_{ m hits}$	Total E_{ABS} , (aJ)	$E_{ABS,av}$, (aJ)
0–92 h (1)	0-50	41	40- 42	1	1.25	92	115	3.95	4120	35.82
	51-200	43	40–48	4	3.92	88	345	3.78	11100	32.17
	201-1993	^a Not	42-54	^a 10	^a 19.3	13	251	0.56	68 700	5284.6
		determined								
Total						193	711	8.29	83920	5353.59
92-42 h (period 2a	0-20	41	40–46	1	1.14	740	841	31.79	28200	38.1
of stage 2)	21-80	42	40–46	2	2.07	590	1222	25.34	40900	69.32
	81-1000	43	40- 55	4	4.76	634	3018	27.23	124000	195.98
	1001-	^a Not	40- 64	^a Not	^a 91.55	92	8423	3.95	224000	2434.7
	10136	determined		determined						
Total						2056	13504	88.31	417100	2738.1
142-168 h (period	0-20	41	40–45	1	1.13	22	25	0.95	917	41.68
2 b of stage 2)	21-601	43	41- 53	3	^a 5.6	43	241	1.84	24300	565.11
Total						65	266	2.79	25217	606.79
168- 183.5 h(3)	0-20	40	40- 42	1	1	8	8	0.34	235	29.37
	21- 100	42	42- 44	2	1.83	6	11	0.25	378	63
Total						14	19	0.59	613	92.37

^a The criteria for identification of an internal mechanism initiated AE [20] are not satisfied.

high energy AE signals ultimately leading to decrease of the overall stress level within the structures [14–18,26]. Then a period of AE 'silence' (denoted in Fig. 2 as a stage 3) followed in the AE history of the conducted experiment. This was also observed during the AE monitoring of the reference OPC sample with encapsulated AI [21].

A visual observation during the experiment confirmed that visible cracks appeared on the surface of the sample as shown in Fig. 3(a). The latter were found also in the cross-section of the structure (Fig. 3(b)) not expected to be formed within the BFS/OPC system at that age of curing [7]. The initiation and/or propagation of these cracks were the very first potential source of AE identified during our test.

A post-test optical microscopy observation on the specimen provided additional evidences for formation of splitting microcracks between the Al rod and the corrosion products layer (Fig. 4(a)). These were formed as a result of the increased compressive forces due to the hydrogen gas generation, friction and release [6,7,26]. Moreover, the presence of microcracks within the corrosion products layer, as shown in Fig. 4(b), was another proof for active micro scale AE sources within the cementitious structure.

In order to differentiate the AE hits recorded from the cement-based structure the analysis was extended from the AE hits rate, providing information for some kind of structural changes, to the parameters of the recorded signals. According to Pollock [9] some of the most commonly used acoustic parameters for characterisation of the AE sources are amplitude, duration, counts and rise time. To the purpose of this study we applied a procedure developed by Wu et al. [20], also described in detail in [21], to classify detected AE waves at each of the stages of the conducted monitoring based on signals' amplitude, duration and counts and associate their appearing with the mechanisms causing damage within the encapsulating composite cement system. The results obtained are summarised in Table 1. Similar parameter-based AE study and differentiation of the recorded acoustic hits extended with calculation of the average ABS energy $(E_{ABS,av})$ as a ratio between the total number of AE hits (N_{hits}) and total ABS energy (E_{ABS}) for the OPC sample with encapsulated Al are reported in [21].

From the data included in Table 1 it can be seen that the AE during stage 1 in Fig. 2 consists of classified AE signals with a short duration (up to $50 \mu s$) and a low amplitude (up to 42 dB) in a good agreement with the data reported for the OPC sample with encapsulated Al for the initial stage of the conducted monitoring [21]. This type of AE can be

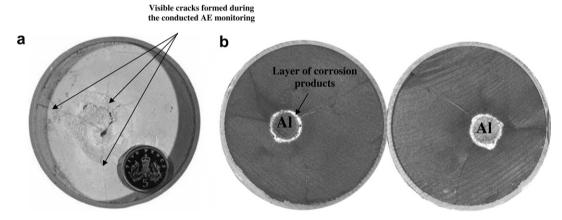


Fig. 3. Visible cracks formed at the (a) surface and (b) cross-section of the BFS/OPC sample with encapsulated Al during the conducted AE monitoring.

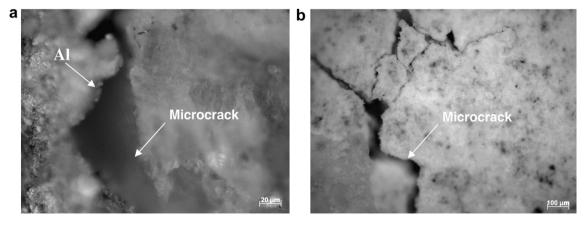


Fig. 4. Micrographs showing (a) splitting crack between the Al rod and corrosion products layer and (b) microcracks within the corrosion products layer found in the BFS/OPC sample with encapsulated Al.

associated with distributed microcracking within the Al corrosion products layer as shown in Fig. 4(b) [17–19]. The group of acoustic signals with similar amplitude and duration detected in a most large population at the beginning of the second stage (period 2a in Fig. 2) can be attributed to localised microcracking within the layer of corrosion products and the onset and/or activation of damage mechanisms such as debonding along the Al rod due to the increase of the compressive forces on the Al/corrosion products layer and corrosion products layer/cement interfaces as a result of hydrogen gas generation and further deposition of corrosion products [8,15–17]. The most energetic hits, however, also recorded in a large number during the period 2a and also 2b in Fig. 2 (Table 1) can be associated with the extension of the splitting cracks between the Al rod and the corrosion products layer (Fig. 4(a)) and the radial crack(s) initiation and/or extension within the cement structure (Fig. 3) caused by the evolution of the hydrogen gas generated due to the Al corrosion [6,7,15– 17,26]. Less than 1% from the total number of the AE hits was recorded during period 3 in Fig. 2. It is important to note that during each of the stages of the experiment there

were AE signals that cannot be classified using the criteria of the procedure from [20]. Moreover, these are characterised by a higher amplitude, a longer duration and $E_{ABS,av}$ in comparison with the classified dominant populations of AE signals (Table 1). The latter is an indication for the activity of different AE sources within the encapsulating cementitious system associated with the observed failure of the structure.

From the analysis performed above and in [21] it is clear that the parameter-based approach provides a good differentiation (in terms of AE) of the mechanisms causing damage in the cementitious samples due to the corrosion of the encapsulated Al. Although the same AE sensor was firmly attached to the samples during the AE experiments and any changes in the time domain parameters of the recorded signals was attributed to the different sources and materials used it is important to consider the strong dependence of the measured acoustic parameters on the experimental setup. For that purpose signal-based approaches were applied using the recorded signal waveforms from the BFS/OPC sample with encapsulated Al to provide a more independent from the experimental setup (sensor coupling

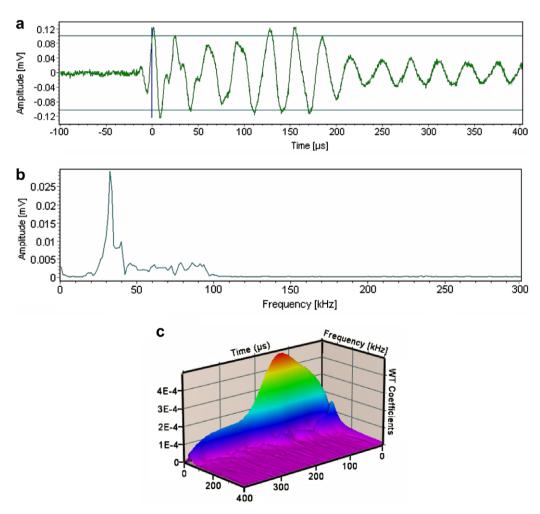


Fig. 5. (a) Typical AE signal waveform, (b) its power spectrum by FFT, and (c) 3D plot of WT detected from the BFS/OPC sample with encapsulated Al during stage 1.

and positioning) results for analysis and conclusions. Conventional fast Fourier transformation (FFT) and wavelet transformation (WT) were used to characterise the processes associated with the Al corrosion and its impact on the mechanical performance of the BFS/OPC structure.

4.2. Relationship between the frequency characteristics of the recorded AE hits and the corrosion of the Al encapsulated in the BFS/OPC sample

Signal-based AE analysis includes advanced mathematical transformations to facilitate the identification of characteristic features of the recorded acoustic waves. Conventional FFT [27] is typically applied to determine the frequency spectra of the detected acoustic waves. The primary frequency, defined as the most energetic frequency component in the Fourier spectrum, has been used to classify different failure modes in concrete and composites [17,28,29]. However, the FFT is of a fixed resolution at all frequencies and can obscure small, high frequency events which usually last for a very short interval of time (a few tens of µs) [29,30]. These events can be characterised by WT. WT is relatively new signal processing technique

applied to AE signal waveforms in order to obtain their frequency-time domain characteristics. Since AE is not a stationary process, i.e., a number of wave modes are present with different frequencies, WT can be helpful to characterise the processes generated AE in heterogeneous cement-based structures as those used in this study.

Continuous WT (CWT) is defined as a set of basic functions obtained by compression/dilation or shifting of a 'mother wavelet'. CWT of a function f(t) in time t is determined as [31]

$$W_{\psi}^{f}(a,b) = \int_{-\infty}^{+\infty} f(t)\psi_{a,b}^{*}(t)\mathrm{d}t \tag{1}$$

where $\psi_{a,b}(t) = \frac{1}{\sqrt{b}} \psi \left(\frac{t-a}{b}\right)$, and $\psi^*(t)$ is complex conjugate of the mother function $\psi(t)$. The two terms, a and b, are known respectively as translation and scale parameters. The translation a defines the shift of the time window through the signal. The scale b in WT corresponds to the scale used in maps, i.e., high scale is used for low frequencies which correspond to global information of the signal and low scale for the frequencies that can provide more details for important mechanical events usually lasting for very short time [17,29–31]. Different mother wavelet

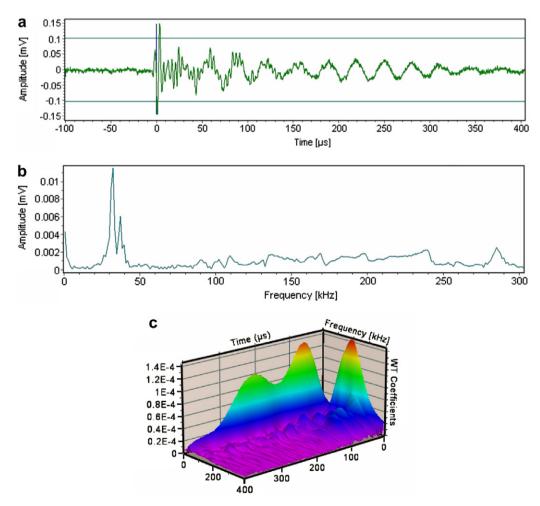


Fig. 6. (a) Typical AE signal waveform, (b) its power spectrum by FFT, and (c) 3D plot of WT detected from the BFS/OPC sample with encapsulated Al during period 2a of stage 2.

functions $\psi(t)$ have been studied and applied for analysis of complex signals such as AE transients. However, the Gabor analysing wavelet was shown as suitable to differentiate failure modes in various materials [29,30]. Gabor mother wavelet $\psi(t)$ is given as [31]

$$\psi(t) = \pi^{-1/4} \left(\frac{\varpi_p}{\gamma}\right)^{1/2} \exp\left[-\frac{t^2}{2} \left(\frac{\varpi_p}{\gamma}\right)^2 + i\varpi_p t\right]$$
 (2)

where ϖ_p is the centre frequency and γ is a constant calculated as $\gamma = \pi (2/\ln 2)^{1/2} = 5.336$.

The WT of the detected acoustic waves from the cementitious wasteform sample with encapsulated Al was calculated and plotted with AGU-Vallen Wavelet freeware [32] based on Gabor mother wavelet with key parameters being: maximum frequency 400 kHz, frequency resolution 1 kHz and wavelet size of 600 samples. The results from WT were performed in 3D plots where x-axis corresponds to the time, y-axis to the frequency and z-axis to the calculated coefficients of $\psi(t)$ for given values of a, b and ϖ_p . For the application of Eq. (1), written for a continuous function f(t), to the sampled acoustic signals a build-in algorithm

was used to limit the integration in the interval between 0 and $N\Delta t$, where N is the number of discrete points (3072 for this study), Δt is the sampling interval and limited range of values given for the shift a and the scale b [31].

Fig. 5 illustrates typical acoustic signal waveform and its frequency spectrum and 3D plot of WT recorded from the BFS/OPC sample with encapsulated Al in a large population (Table 1) during stage 1 in Fig. 2. This type of resonance-like signals with a primary frequency at 34 kHz (Fig. 5(b)) shows that the low frequency components were clearly transmitted by the sensor even though characterised by a relatively low sensitivity for this range of frequencies. Therefore there was negligible effect of the sensor coupling or ringing effect caused by the frequency response of the sensor on the recorded AE waves. Moreover, from the plot in Fig. 5(b) it can be seen that this AE signal is also characterised by intensity in the Fourier spectrum up to 100 kHz. The 3D plot of WT (Fig. 5(c)) shows that these two main frequency components: below and above 40 kHz constituent the first 100 µs of the signal waveform and can be directly related to the initial features of the AE sources [33]. As stated by Grosse et al. [33] after a

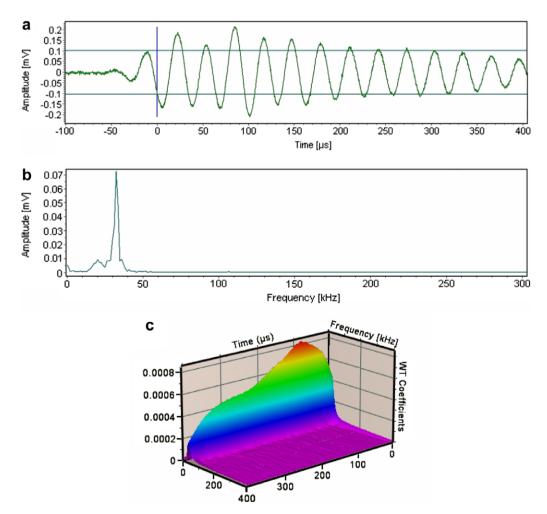


Fig. 7. (a) Typical AE signal waveform, (b) its power spectrum by FFT, and (c) 3D plot of WT detected from the BFS/OPC sample with encapsulated Al during period 2b of stage 2.

few oscillations the signals generated by the sensor are dominated by side reflections or geometry, heterogeneity of the materials under study, the intrinsic properties of the sensor such as resonant frequency and sensor coupling. The presence of those frequency components in the Fourier spectrum of the signal waveform in Fig. 5(a) is an indication for the complex nature of the AE originating from

the Al corrosion process associated with accumulation of corrosion products and hydrogen gas generation and release causing cracking of the encapsulating cementitious structure.

Similar separation of the frequency components as dominant low (below 40 kHz) and high (above 40 kHz up to 300 kHz) was found in the Fourier spectrum of the acoustic

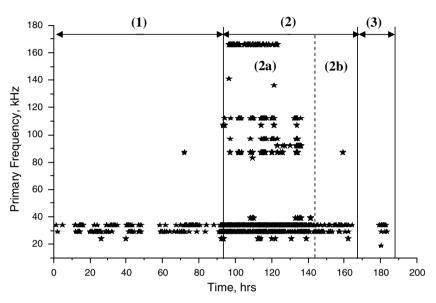


Fig. 8. Primary frequency of the AE signals detected from the BFS/OPC sample with encapsulated Al.

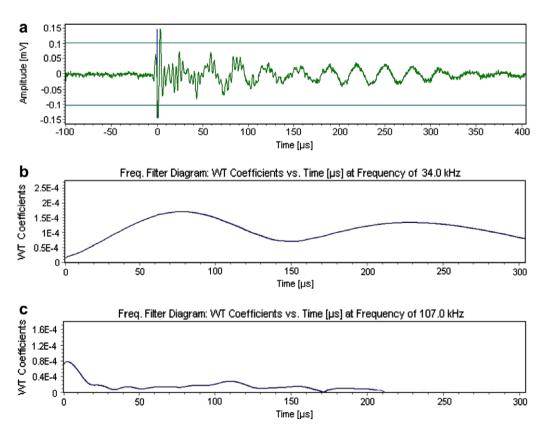


Fig. 9. (a) Typical AE signal waveform and its 2D plots of WT at frequencies of (b) 34 kHz and (c) 107 kHz recorded from the BFS/OPC sample with encapsulated Al during period 2a of stage 2.

signal waveform recorded during period 2a from stage 2 (Fig. 2) as shown in Fig. 6(b). Again these frequency modes arrived within the first 100 μs of the duration of the AE hit (Fig. 6(c)). Therefore similar processes as those identified during stage 1 could be responsible for the generation of these acoustic waves suggested to be localised microcracking and debonding between the Al rod and the corrosion products layer.

For the most energetic AE hits recorded during the periods denoted as 2a and 2b in Fig. 2 the frequency spectrum was dominated by the low frequency component at 34 kHz (Fig. 7(b)). Visible cracks formation and/or extension within the cement structures along with hydrogen gas evolution could be associated with the generation and release of this type of acoustic waves.

According to their primary frequency all detected AE hits from the BFS/OPC sample with encapsulated Al throughout the duration of the experiment were clustered for frequencies below 40 kHz, between 90 and 110 kHz and around 170 kHz as shown in Fig. 8. Similar results were obtained for the primary frequency of the AE hits detected from the OPC sample with encapsulated Al [23].

The WT analysis can be also used to provide information on the time of occurrence and the duration of the low and high frequency components identified within the acoustic signal waveforms [29]. 2D plots of WT at frequencies of 34 kHz and 107 kHz are illustrated in Figs 9 and 10 for the AE hits detected from the BFS/OPC sample with

encapsulated Al respectively during period 2a and 2b of stage 2 in Fig. 2.

The dominant low frequency component at 34 kHz, as shown in Figs 9(b) and 10(b), lasted during the first 100 µs whereas the selected high frequency wave mode at 107 kHz was present for less than 20 µs of the duration of the hit. Moreover, the high frequency component arrived first; hence this very short event from the AE hit can be associated with microcracks formation and propagation. The AE sources characterised by the dominant low frequency component of the signal waveforms can be related to the macro scale damage development within the structure due to the formation and/or propagation of visible cracks and hydrogen gas evolution.

5. Discussion

A large number of 2328 AE hits were recorded from the BFS/OPC sample with encapsulated Al for 183.5 h of monitoring separated into clearly defined periods with very low or almost none activity followed by an abrupt jump in the AE hits number. This behavior of the sample, in terms of AE, resembled the response of studied cement-based structures under loading associated with micro and macro scale damage development [14–20]. Moreover, the interval of AE 'silence' following the period of a high AE activity in the history of the conducted experiment (stage 2 in Fig. 2)

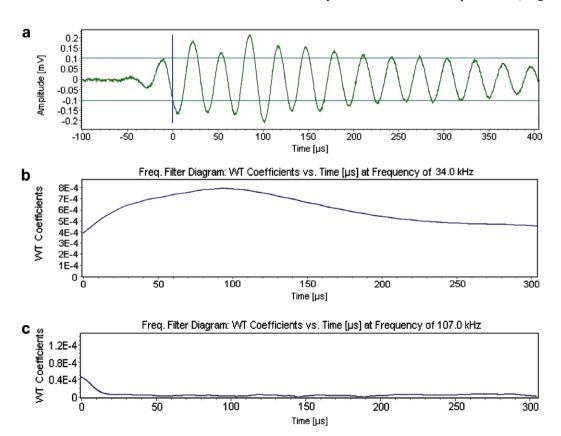


Fig. 10. (a) Typical AE signal waveform (a) and its 2D plots of WT respectively at frequencies of (b) 34 kHz and (c) 107 kHz recorded from the BFS/OPC sample with encapsulated Al during period 2b of stage 2.

showed Kaiser effect observed in cement-based materials during loading-unloading cycles [18].

The applied conventional AE parameter-based analysis, following the procedure from Wu et al. [20], allowed to classify the AE hits according to their duration, amplitude and counts number. The results obtained provided the first point to conclude the complex nature (in terms of AE) and activity of a large number of processes (sources) later confirmed by visual and microscopy observations.

According to their parameters the AE hits recorded from the BFS/OPC sample with encapsulated Al, as well as for the signals recorded from the OPC structure reported in [20,21], can be generally divided in two main groups. The first group consists of a large number of signals with a duration up to 80 µs and an amplitude up to 46 dB associated with microcracks initiation and propagation. Moreover, the second smaller group of signals, with an amplitude up to 55 dB and a duration more than 80 µs, were associated with critical size crack(s) formation and/ or extension and hydrogen gas release within the cementitious structure with encapsulated Al. The results from the FFT and WT analysis revealed that the AE events detected from the BFS/OPC structures with encapsulated Al consist of two main frequency components: low (primary frequency at 34 kHz) and high (above 40 kHz). Therefore it can be suggested that two mechanisms of AE co-exist together associated with microcracks formation and localization in a fracture process zone ultimately leading to visible cracks initiation and propagation. Very similar results were obtained from the AE data analysis performed on a reference OPC sample with encapsulated Al as reported in [21–23].

It is worth noting that under very similar experimental conditions, preparation and age of curing the AE hits recorded were significantly less for the BFS/OPC sample in comparison with that for the OPC system [21]. The latter is in a good agreement with the reported by Setiadi [6,7] lower rate of Al corrosion in composite cements than in pure OPC. However, the basic parameters and frequency characteristics of the AE signals did not reveal any important differences; thus the sources and the propagation properties of the structures (in terms of AE) are similar under the established experimental conditions. Additional evidence for the decreased rate of the Al corrosion and therefore the less damage developed within the BFS/OPC sample prior to the conducted test is also the intensity of the AE generation and release calculated for the stages 2 in Fig. 2 for the BFS/OPC sample to be 41.1 AE hits/hour whereas for the OPC structure during the similar period of a high AE activity it was reported to be 335 AE hits/hour [21].

6. Conclusions

The AE technique and methods for analysis of the collected data were shown feasible for monitoring and assessment of the mechanical performance of cementitious

structures with encapsulated metallic wastes such as Al. The methodology of analysis was based on AE processing accounting for hits rate, signal parameters and frequency characteristics along with visual and microscopy observations which enabled identification of major AE sources in the simulant cementitious wasteforms. Distributed and localised microcracking, hydrogen gas evolution and visible cracks formation and extension within the cementitious matrix can be differentiated by their AE response in terms of signals duration, amplitude and frequency components. Despite of some differences found in the AE hits rate and their time distribution detected from the BFS/OPC sample with encapsulated Al in comparison with those from the reference OPC structure under identical experimental conditions, the AE signals detected have been successfully classified based on similar parameters and frequency characteristics. To evaluate the feasibility of the method for industrial use an optimization of the experimental setup will be required as well as additional research to assess the importance of various parameters such as sensor frequency characteristics, positioning and coupling, background noise, waste type and loading, container volume and shape.

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